

# Nanostructures of Polyaniline Doped with Inorganic Acids

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**ABSTRACT:** With an average diameter of 150–340 nm and a conductivity of  $10^{-1}$ – $10^0$  S/cm, nanostructures (e.g., nanotubes or nanorods) of polyaniline (PANI) were synthesized by a self-assembly method in the presence of inorganic acids (e.g., HCl, H<sub>2</sub>SO<sub>4</sub>, HBF<sub>4</sub>, and H<sub>3</sub>PO<sub>4</sub>) as dopants. It was found that the morphology, size, and electrical properties of the resulting nanostructures depended on the dopant structures and the reaction conditions. In particular, all these PANI nanostructures showed hydrophilic features, and the contact angles with water were measured to be about 27–40° depending on the dopant. The FTIR spectrum, UV–vis absorption spectrum, XPS, and X-ray diffraction were used to characterize the molecular structures of the nanostructures. It was found that their main chain structure and electric structure were identical to those of the emeraldine salt form of PANI. The micelles formed by anilinium cations act as the template like in the formation of PANI nanostructures.

## Introduction

Studies on molecular wire have attracted more and more attention since the discovery of the first molecular wire of carbon nanotubes (CNTs);<sup>1</sup> thus, conducting polymers are promising in this area due to their long conjugated length and metallike conductivity.<sup>2,3</sup> Nanotubes or nanofibers of conducting polymers, such as polyacetylene,<sup>4</sup> poly(3-methylthiophene),<sup>5</sup> polypyrrole,<sup>6</sup> and polyaniline,<sup>7</sup> have been synthesized by a “template synthesis” method. In our laboratory, Wan et al.<sup>8</sup> reported that microtubes of polyaniline (PANI) and polypyrrole (PPy) were synthesized by in-situ doping polymerization in the presence of  $\beta$ -naphthalenesulfonic acid ( $\beta$ -NSA) as a dopant. Here,  $\beta$ -NSA acted as a template like in the formation of PANI-( $\beta$ -NSA) microtubules due to its surfactant function (e.g., –SO<sub>3</sub>H group). However, it was different from the “template synthesis” method. Unlike template synthesis,  $\beta$ -NSA did not need to be removed after polymerization because it can act as a dopant for PANI at the same time. Therefore, we call it a self-assembly method. Moreover, we have proved the reliability and practicability of this self-assembly method for synthesizing microtubes or nanotubes of conducting polymers through changing polymer chains<sup>9–14</sup> and dopants<sup>15–17</sup> and using different polymerization methods.<sup>18</sup> The dopants used in our previous experiments have both doping and surfactant functions. The surfactant function of the dopant seemed to play an important role in the formation of nanostructures of conducting polymers.<sup>19</sup> So, it is very interesting to examine whether the surfactant function of the dopant is the prerequisite to form self-assembled PANI nanostructures.

In this article, self-assembled PANI nanostructures (e.g., nanotubes or nanorods) with an average diameter of 150–340 nm and a room-temperature conductivity of  $10^{-1}$ – $10^0$  S/cm in the presence of inorganic acids (e.g., HCl, H<sub>2</sub>SO<sub>4</sub>, H<sub>3</sub>PO<sub>4</sub>, and HBF<sub>4</sub>) as dopants are reported for the first time. The effect of the dopant structures and reaction conditions on the morphology, size, and physical properties of the resulting PANI nanostructures was investigated, and the formation mechanism of PANI nanostructures was discussed.

## Experimental Section

Aniline monomer was distilled under reduced pressure. Other reagents, such as dopants (HCl, H<sub>2</sub>SO<sub>4</sub>, HBF<sub>4</sub>, and H<sub>3</sub>PO<sub>4</sub>), oxidant (ammonium peroxydisulfate, APS), sodium dodecylbenzenesulfonate (SDBS), and hexadecyltrimethylammonium bromide (HTAB) as surfactants, were used as received.

The nanostructures of PANI doped with inorganic acids were synthesized by a self-assembly method. To understand the effect of the surfactant on the morphology of PANI, the parallel experiments with and without a surfactant were carried out. For example, PANI doped with H<sub>3</sub>PO<sub>4</sub> was synthesized as follows: Aniline (0.2 mL) and SDBS (3.3 mg) were mixed with H<sub>3</sub>PO<sub>4</sub> (0.06 mL) in 10 mL of deionized water in the ice bath to form a white dispersion of aniline/H<sub>3</sub>PO<sub>4</sub> salt. Then an aqueous solution of APS (0.46 g in 5 mL of deionized water) was added to the above mixtures. The polymerization was carried out for 12 h in the ice bath. A green solid of PANI–H<sub>3</sub>PO<sub>4</sub> was obtained after rinsing with H<sub>2</sub>O, CH<sub>3</sub>OH, and CH<sub>3</sub>OCH<sub>3</sub> for three times. PANI–HCl, PANI–H<sub>2</sub>SO<sub>4</sub>, and PANI–HBF<sub>4</sub> were synthesized by a similar method. Meanwhile, PANI–H<sub>3</sub>PO<sub>4</sub> was also synthesized without a surfactant through the same process for a comparative study.

The UV–vis spectrum, FTIR spectrum, X-ray photoelectron spectroscopy (XPS), and X-ray diffraction were used to characterize the molecular structures of the resulting PANI nanostructures. FTIR spectra were performed on a Bruker EQUINOX55, while X-ray diffraction was carried on a RINT 2000 Wilder-angle goniometer. The morphology was measured by a scanning electron microscope (SEM, Hitachi-530 or FESEM, JSM-6700F) and a transmission electron microscope (TEM, Hitachi-9000). The conductivity at room temperature was measured by a Keithley 196 SYSTEM DM digital multimeter and an ADVANTEST R6142 programmable dc voltage/current source, a standard four-probe method. XPS was performed on ES-300 (Kratos). The contact angle with water for the nanostructural film deposited on the glass substrate was measured on a contact angle system (OCA Dataphysics DCAT 11).

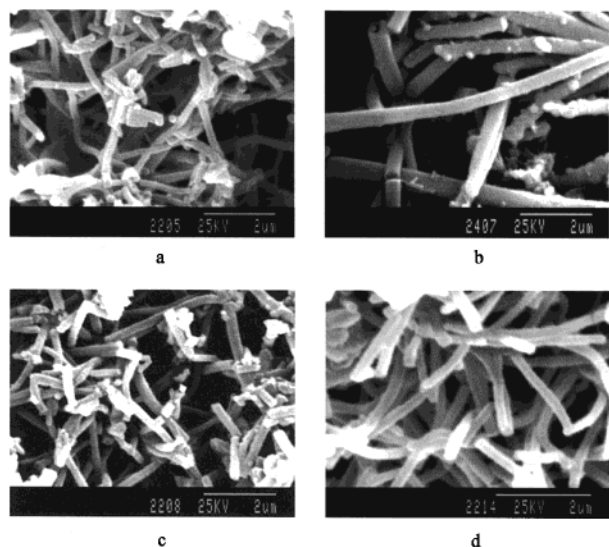
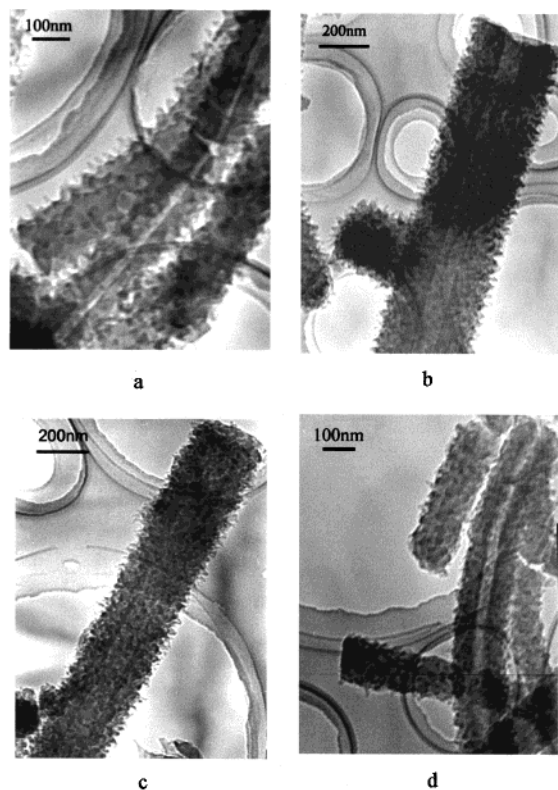
## Results and Discussion

**Morphologies of PANI and Their Formation Mechanism.** As shown in Figure 1, all PANI samples synthesized in the presence of SDBS as a surfactant

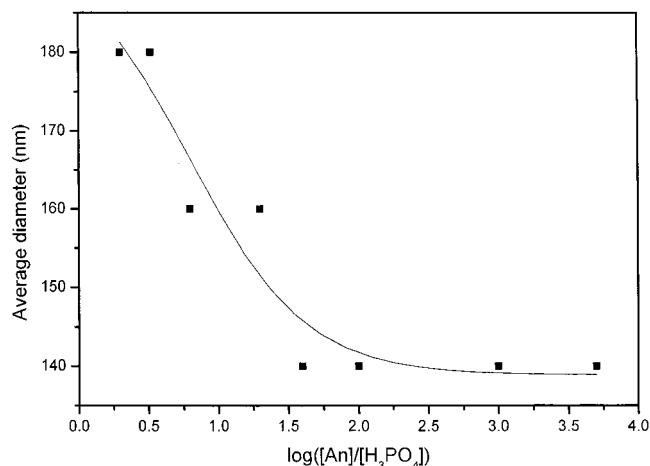
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**Table 1. Detailed Data of PANI Nanostructures Doped with Different Inorganic Acids**

	PANI-HCl	PANI-H <sub>2</sub> SO <sub>4</sub>	PANI-HBF <sub>4</sub>	PANI-H <sub>3</sub> PO <sub>4</sub>
$\sigma_{RT}$ (S/cm)	6.38	1.26	1.12	0.49
[N <sup>+</sup> ]/[N] (%)	13.5	14.1	14.9	18.3
average diameter (nm)	150	335	265	180
$T_0$ (K)	$6.60 \times 10^3$	$8.96 \times 10^3$	$1.14 \times 10^4$	$1.51 \times 10^4$
UV-vis absorption $\lambda_{max}$ (nm)	423, 890	423, 881	430, 877	452, 833

**Figure 1.** SEM images of PANI nanostructures doped with different inorganic acids: (a) HCl, (b) H<sub>2</sub>SO<sub>4</sub>, (c) HBF<sub>4</sub>, (d) H<sub>3</sub>PO<sub>4</sub> ([An]/[acid] = 2, [SDBS] =  $2.4 \times 10^{-3}$  M, at 0–4 °C).**Figure 2.** TEM images of PANI nanostructures doped with different inorganic acids: (a) HCl, (b) H<sub>2</sub>SO<sub>4</sub>, (c) HBF<sub>4</sub>, (d) H<sub>3</sub>PO<sub>4</sub> ([An]/[acid] = 2, [SDBS] =  $2.4 \times 10^{-3}$  M, at 0–4 °C).

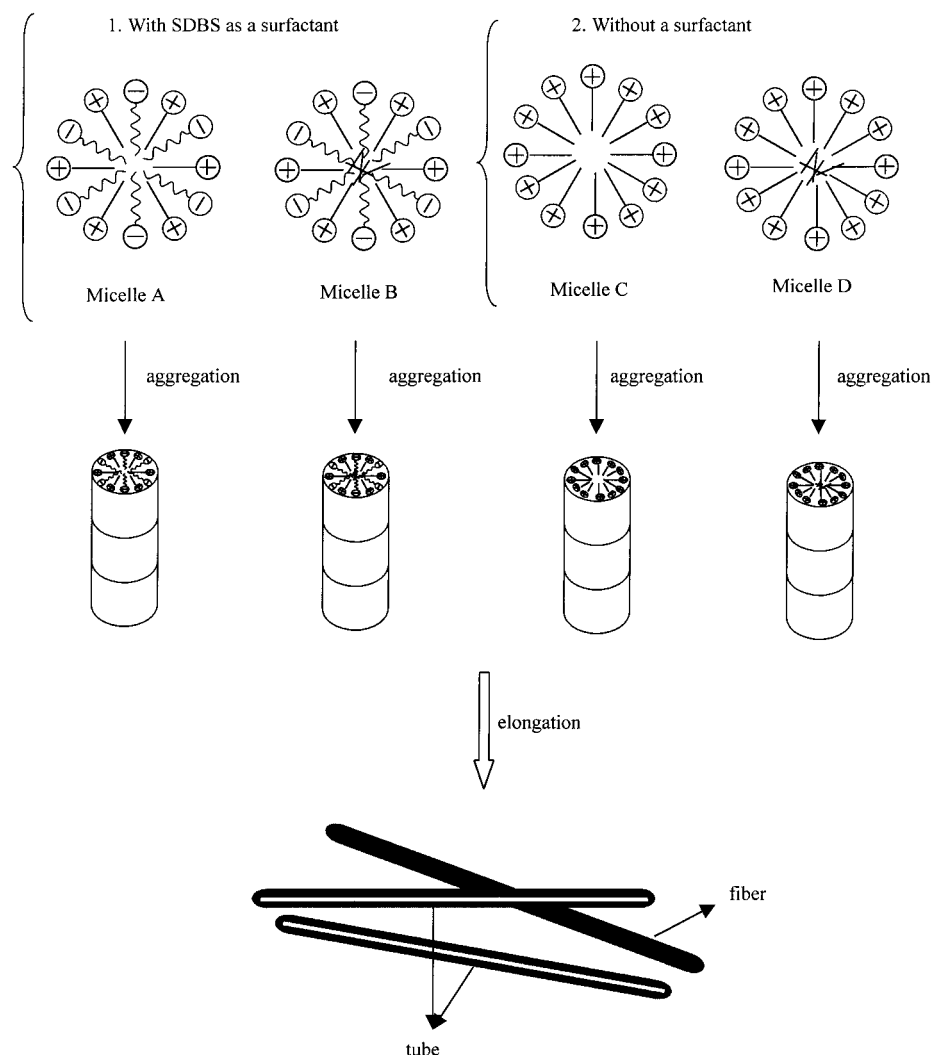
have fibrous morphology. TEM images reveal that these fibers include tubes and rods as shown in Figure 2. It was found that the size and formation probability of the nanostructures strongly depended on the dopant structure and reaction conditions (e.g., aniline/acid molar

**Figure 3.** Influence of [An]/[H<sub>3</sub>PO<sub>4</sub>] on the average diameter of PANI-H<sub>3</sub>PO<sub>4</sub> ([SDBS] =  $2.4 \times 10^{-3}$  M, at 0–4 °C).

ratio and reaction temperature). For instance, the diameters of PANI nanostructures varied from 150 to 340 nm, depending on the dopant used (Table 1). The result indicates that the size of PANI nanostructures is controllable by changing the dopant structure. Moreover, the size of PANI nanostructures was affected by the aniline/acid mole ratio (represented by [An]/[acid]). Taking PANI-H<sub>3</sub>PO<sub>4</sub> as an example, the average diameter of PANI-H<sub>3</sub>PO<sub>4</sub> decreased slightly from 180 to 140 nm when [An]/[H<sub>3</sub>PO<sub>4</sub>] changed from 1:0.5 to 1:0.01. However, as shown in Figure 3, no changes took place in the average diameter of PANI-H<sub>3</sub>PO<sub>4</sub> while [An]/[H<sub>3</sub>PO<sub>4</sub>] changed from 1:0.01 to 1:0.0002.

The formation probability of PANI nanostructures depended on reaction temperature and [An]/[acid]. It was found that the reaction temperature had an important influence on the morphology of the resulting PANI. Taking PANI-HCl as an example, besides the fibrous nanostructures, there were some granular solids when aniline was polymerized at room temperature ([An]/[acid] = 1:0.5). When the polymerization occurred at a lower temperature (0–4 °C) as mentioned above, nearly all the products showed fibrous nanostructures, and the formation probability of the nanotubes or rods was greatly enhanced. A similar result was observed when H<sub>2</sub>SO<sub>4</sub> was used as the dopant. In addition, the ratio [An]/[acid] also affected the formation probability of fibrous nanostructures greatly when the polymerization was carried out at a low temperature (0–4 °C). PANI-H<sub>3</sub>PO<sub>4</sub>, for example, showed granular morphology when [An]/[H<sub>3</sub>PO<sub>4</sub>] was 1:2 or 1:3. However, some aniline/H<sub>3</sub>PO<sub>4</sub> mole ratios (e.g., 1:0.5, 1:0.3, and 1:0.16) were more favorable for the formation of fibrous PANI-H<sub>3</sub>PO<sub>4</sub> nanostructures. Then the formation probability of the nanostructures decreased with the increase of [An]/[H<sub>3</sub>PO<sub>4</sub>]. It needs to be pointed out that only a small quantity of nanostructures was obtained when [An]/[H<sub>3</sub>PO<sub>4</sub>] reached 1:0.0002.

The influence of the surfactant structure and concentration on the morphology of PANI nanostructures was



**Figure 4.** Schematic diagram of the formation mechanism for PANI nanostructures synthesized by a self-assembly process.

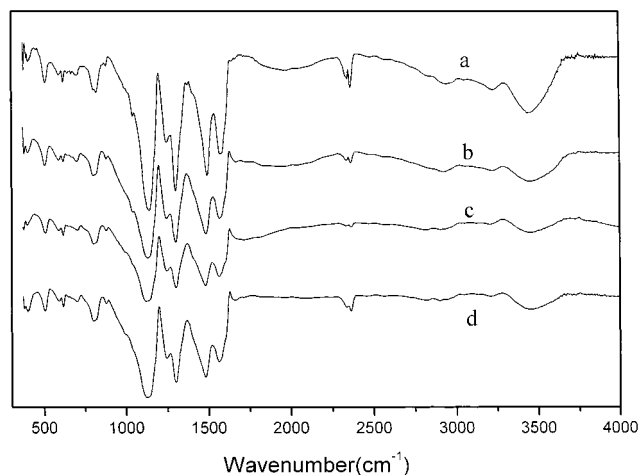
investigated in order to examine whether the surfactant is the prerequisite to form self-assembled PANI nanostructures. It was found that PANI- $\text{H}_3\text{PO}_4$  had the same diameter ( $\sim 180$  nm) when the concentration of SDBS changed from  $2.4 \times 10^{-3}$  to  $1.3 \times 10^{-2}$  M, showing the concentration of SDBS had little effect on the morphology and size of PANI- $\text{H}_3\text{PO}_4$  nanostructures. When using HTAB instead of SDBS as the surfactant, the average diameter of PANI- $\text{H}_3\text{PO}_4$  nanostructures only varied from 180 to 200 nm with the concentration of HTAB changing from  $8.0 \times 10^{-3}$  to  $1.6 \times 10^{-2}$  M. And then the diameter kept constant even when the concentration of HTAB increased to  $5 \times 10^{-2}$  M. This result is consistent with that of PANI- $\text{H}_3\text{PO}_4$  synthesized in the presence of SDBS as a surfactant. PANI- $\text{H}_3\text{PO}_4$  nanostructures, with their diameter of 150 nm, were also obtained even though no surfactant was added. Thus, the above results indicate that surfactant function of the dopant is not the prerequisite for the formation of self-assembled PANI nanostructures. However, the addition of the surfactant and its concentration do affect the size of PANI nanostructures.

How does one interpret the formation of PANI nanostructures by a self-assembly process?

If SDBS is used as the surfactant (represented by  $\phi\sim$ ), it is easy to form micelles. At the same time, aniline may exist in the form of anilinium salt (represented by  $\oplus-$ ) or free aniline (represented by

in the reaction solution. Spherical micelles as templates may be formed.<sup>20</sup> Anilinium cations can be solubilized in the micelle-water interface to form micelle A, or a part of free aniline diffuses into micelles to form micelle B<sup>21</sup> as shown in Figure 4. Micelles A and B are assumed as the templates to form PANI nanostructures in the presence of a surfactant. On the other hand, it is expected that micelles C and D (Figure 4) formed by anilinium cations might be the templates to form PANI nanostructures in the absence of a surfactant. Solubilized aniline or anilinium molecules are polymerized oxidatively by APS existing in the aqueous phase. The reaction takes place mainly in the micelle-water interface adjacent to the surfactant headgroups because hydrated APS molecules cannot penetrate into the micelle surface.<sup>22</sup> With the polymerization proceeding, the micelles become big spheres through accretion<sup>23</sup> or tubes/rods through elongation<sup>24</sup> depending on the local conditions. In our experiments, it is obvious that the occurrence of the elongation procedure judges from the morphology of the resulting PANI. Then micelles A or C will be converted into tubes and micelles B or D into rods. The size of the micelles may greatly affect the size of the resulting nanostructures; therefore, PANI samples doped with different inorganic acids have different diameters. Up to now, it is reasonable to conclude that PANI nanostructures doped with inorganic acids without any surfactant function can be





**Figure 5.** FTIR spectra of PANI nanostructures doped with different inorganic acids: (a)  $\text{H}_2\text{SO}_4$ , (b)  $\text{HCl}$ , (c)  $\text{HBF}_4$ , (d)  $\text{H}_3\text{PO}_4$  ( $[\text{An}]/[\text{acid}] = 2$ ,  $[\text{SDBS}] = 2.4 \times 10^{-3} \text{ M}$ , at  $0-4^\circ\text{C}$ ).

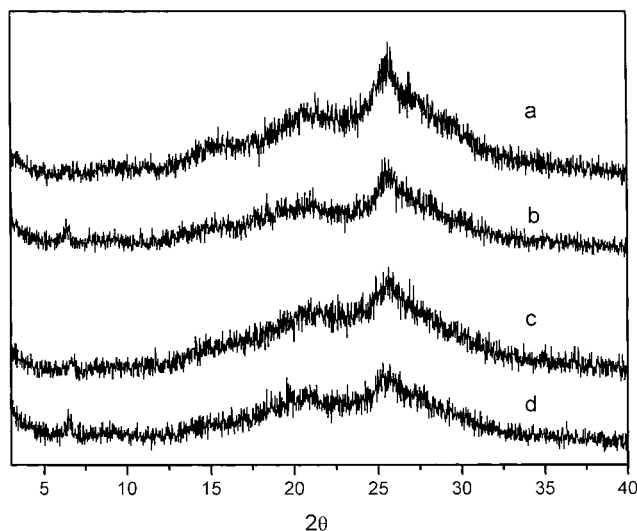
synthesized by a self-assembly process. However, the diameter of PANI nanostructures can be affected through the addition of the surfactant and change in its concentration.

**Structural Characterization.** The UV-vis spectrum, FTIR spectrum, and X-ray diffraction were used to characterize the molecular structure of the resulting nanostructures of PANI doped with different inorganic acids.

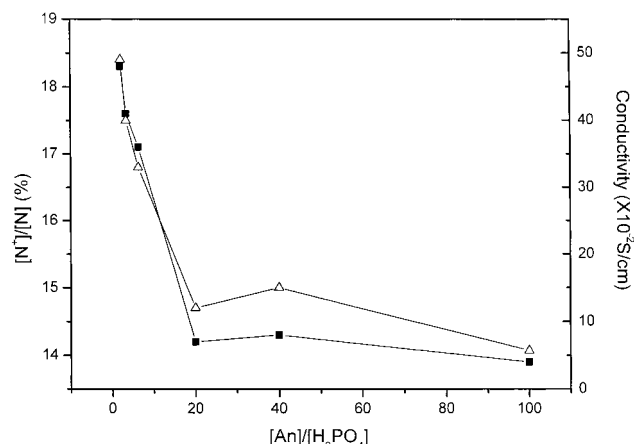
Figure 5 shows the FTIR spectra of PANI-HCl, PANI- $\text{H}_2\text{SO}_4$ , PANI- $\text{HBF}_4$ , and PANI- $\text{H}_3\text{PO}_4$  nanostructures prepared in the presence of SDBS as a surfactant. It was found that the FTIR spectra of these four PANI were similar. The characteristic peaks at 1567 and  $1483 \text{ cm}^{-1}$  can be assigned to the stretching vibration of quinoid ring and benzenoid ring, respectively. The bands at 1300 and  $1246 \text{ cm}^{-1}$  correspond to C-H stretching vibration with aromatic conjugation. These characteristic peaks are identical to those of PANI-( $\beta$ -NSA) microtubes<sup>25</sup> and those of PANI-HCl<sup>26</sup> prepared in a common method. The FTIR spectrum of PANI- $\text{H}_3\text{PO}_4$  prepared without surfactant is the same as that of PANI- $\text{H}_3\text{PO}_4$  prepared with SDBS acting as the surfactant. The results indicate that the backbone structures of PANI-HCl, PANI- $\text{H}_2\text{SO}_4$ , PANI- $\text{HBF}_4$ , and PANI- $\text{H}_3\text{PO}_4$  nanostructures obtained in this work are identical to each other and also to those of PANI-( $\beta$ -NSA) microtubes<sup>25</sup> and PANI-HCl<sup>26</sup> reported previously.

UV-vis spectra of the resulting PANI nanostructures dissolved in *m*-cresol were measured. For all these four PANI nanostructures, two bands at about 430 nm and longer than 850 nm with a long tail are present (Table 1). The peak at longer than 850 nm can be assigned to a polaron band.<sup>27</sup> Table 1 shows that the electrical structures of PANI nanostructures are slightly different, depending on the dopants used. However, all these PANI nanostructures are identical to the emeraldine salt form of PANI.<sup>28</sup>

X-ray scattering patterns of PANI-HCl, PANI- $\text{H}_2\text{SO}_4$ , PANI- $\text{HBF}_4$ , and PANI- $\text{H}_3\text{PO}_4$  nanostructures are shown in Figure 6. Two broad peaks centered at  $2\theta = 20^\circ$  and  $26^\circ$  were observed, showing the resulting PANI nanostructures are amorphous. The peak centered at  $2\theta = 20^\circ$  may be ascribed to periodicity parallel to the polymer chain, while the latter peaks may be caused by the periodicity perpendicular to the polymer chain.<sup>29</sup>



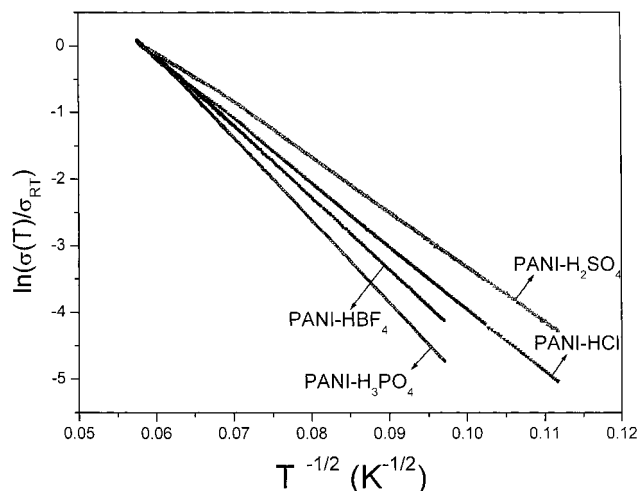
**Figure 6.** X-ray diffraction of PANI nanostructures doped with different inorganic acids: (a)  $\text{H}_2\text{SO}_4$ , (b)  $\text{HCl}$ , (c)  $\text{HBF}_4$ , (d)  $\text{H}_3\text{PO}_4$  ( $[\text{An}]/[\text{acid}] = 2$ ,  $[\text{SDBS}] = 2.4 \times 10^{-3} \text{ M}$ , at  $0-4^\circ\text{C}$ ).



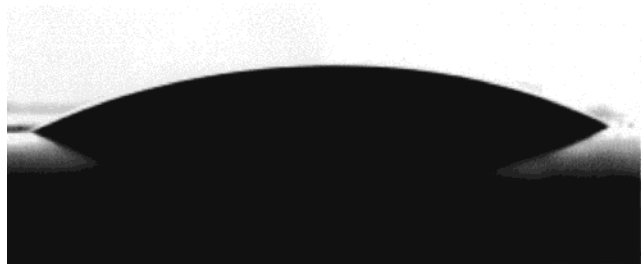
**Figure 7.** Effect of  $[\text{An}]/[\text{H}_3\text{PO}_4]$  on room temperature conductivity and doping degree of PANI nanostructures:  $\triangle$ , room temperature conductivity;  $\blacksquare$ , doping degree ( $[\text{An}]/[\text{H}_3\text{PO}_4] = 2$ ,  $[\text{SDBS}] = 2.4 \times 10^{-3} \text{ M}$ , at  $0-4^\circ\text{C}$ ).

However, it is noted that the crystallinity of PANI nanostructures doped with inorganic acids is lower than that of PANI-( $\beta$ -NSA).<sup>30</sup> This may be due to different molecular sizes of the dopants or different formation mechanisms of the nanostructures.

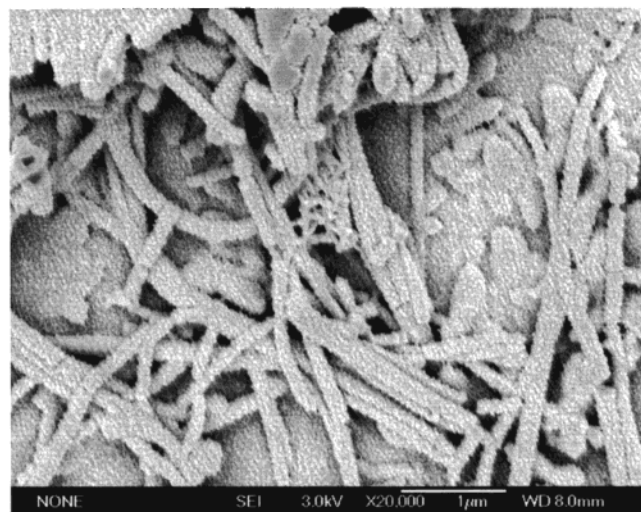
**Physical Properties.** PANI doped with HCl,  $\text{H}_2\text{SO}_4$ ,  $\text{HBF}_4$ , and  $\text{H}_3\text{PO}_4$  has different room-temperature conductivity within the range of  $0.49-6.38 \text{ S/cm}$ , depending on the dopant used. It was noted that their conductivity were 10–100 times higher than that of PANI nanotubes doped with  $\beta$ -NSA ( $\sim 10^{-2} \text{ S/cm}$ )<sup>30</sup> prepared at an aniline/NSA mole ratio of 1:0.5 ( $0-4^\circ\text{C}$ ). An order of PANI-HCl > PANI- $\text{H}_2\text{SO}_4$  > PANI- $\text{HBF}_4$  > PANI- $\text{H}_3\text{PO}_4$  in conductivity was observed, and the detailed data are provided in Table 1. The conductivity of PANI- $\text{H}_2\text{SO}_4$  and PANI- $\text{HBF}_4$  is consistent with the results of doping degree, assigned as  $[\text{N}^+]/[\text{N}]$  measured by XPS. However, the conductivity of PANI-HCl and PANI- $\text{H}_3\text{PO}_4$  is not consistent with XPS results (see Table 1). A possible reason may be that the molecular size of  $\text{H}_3\text{PO}_4$ ,<sup>31</sup> which is larger than that of HCl, resulted in its lower conductivity. This is consistent with  $T_0$  values as shown in Table 1.



**Figure 8.** Temperature dependence of conductivity for PANI nanostructures doped with different inorganic acids ([An]/[acid] = 1:0.5, [SDBS] =  $2.4 \times 10^{-3}$  M, at 0–4 °C).



a



b

**Figure 9.** Photograph of water drop on PANI–H<sub>3</sub>PO<sub>4</sub> film deposited on the glass substrate (a) and SEM of PANI–H<sub>3</sub>PO<sub>4</sub> film (b) ([An]/[acid] = 2, [SDBS] =  $2.4 \times 10^{-3}$  M, at 0–4 °C).

The effect of [An]/[H<sub>3</sub>PO<sub>4</sub>] on the conductivity of PANI–H<sub>3</sub>PO<sub>4</sub> nanostructures was measured. However, there was no significant difference in conductivity, around  $10^{-2}$  S/cm, when [An]/[H<sub>3</sub>PO<sub>4</sub>] varied from 1:0.5 to  $1:2.5 \times 10^{-2}$ . As the aniline/H<sub>3</sub>PO<sub>4</sub> mole ratio reached  $1:1.0 \times 10^{-3}$ , the conductivity decreased to  $5.7 \times 10^{-2}$  S/cm. This result is consistent with the doping degree as shown in Figure 7.

The temperature dependence of conductivity of PANI doped with different inorganic acids was measured

between 50 and 300 K. It was found that the conductivity of all measured samples decreased with decreasing temperature, exhibiting a typical semiconductor behavior. The data are best fit to the relationship of  $\ln \sigma$  plotted vs  $T^{-1/2}$  as shown in Figure 8. So it is reasonable to believe that the temperature dependence of conductivity of all these PANI nanostructures is in agreement with one-dimensional variable range hopping (1D-VRH) model proposed by Mott,<sup>32</sup> which is expressed as

$$\sigma(T) = \sigma_0 \exp[-(T_0/T)^{1/n+1}], \quad n = 1, 2, 3$$

where  $\sigma_0$  is a constant,  $T_0$  the hopping barrier, and  $T$  the Kelvin temperature. The  $T_0$  values of PANI nanostructures doped with different acids are comparable to those reported in the literature.<sup>33</sup> The  $T_0$  value follows an order of PANI–H<sub>3</sub>PO<sub>4</sub> > PANI–HBF<sub>4</sub> > PANI–H<sub>2</sub>SO<sub>4</sub> > PANI–HCl, which is consistent with a trend of their room-temperature conductivity.

Interestingly, it was found that PANI–H<sub>3</sub>PO<sub>4</sub> nanostructural films deposited on the glass substrate exhibited hydrophilic behavior, which was confirmed by measuring contact angle with water ( $\theta = 32^\circ$ ). Figure 9 shows the photograph of water drop on PANI–H<sub>3</sub>PO<sub>4</sub> film (a) and SEM of PANI–H<sub>3</sub>PO<sub>4</sub> nanostructure film (b). Similarly, PANI–HCl, PANI–H<sub>2</sub>SO<sub>4</sub>, and PANI–HBF<sub>4</sub> films all are hydrophilic, and their contact angles with water are  $40^\circ$ ,  $38^\circ$ , and  $27^\circ$ , respectively.

## Conclusions

Nanostructures (e.g., nanotubes or nanorods) of PANI doped with HCl, H<sub>2</sub>SO<sub>4</sub>, HBF<sub>4</sub>, and H<sub>3</sub>PO<sub>4</sub> with an average diameter of 150–340 nm and a conductivity of  $10^{-1}$ – $10^0$  S/cm were synthesized in a self-assembly method with and without a surfactant. It was found that the morphology, size, room-temperature conductivity, and  $T_0$  value of the nanostructures depended on the dopant structures and reaction conditions. In particular, lower reaction temperature (e.g., 0–4 °C) and some [An]/[acid] ratio values (1:0.5, 1:0.3, and 1:0.16) were favorable to the formation of PANI nanostructures. In the presence of a surfactant, micelles formed by anilinium cations and surfactant anions were regarded as templates in the formation of the nanostructures. In the absence of a surfactant, on the other hand, micelles formed by anilinium cations were considered as templates. However, the size of PANI nanostructures was slightly affected by the addition of the surfactant during the polymerization. Interestingly, the resulting PANI nanostructural films show hydrophilic behaviors, and the contact angles with water depend on the dopant used. FTIR and UV–vis spectra, XPS, and X-ray diffraction measurements show that the main chain structure and electrical structure of PANI nanostructures are identical to those of the emeraldine salt form of PANI. Moreover, the resulting nanostructures are also amorphous, similar to granular PANI doped with inorganic acids.

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## References and Notes

- (1) Iijima, S. *Nature (London)* **1991**, 354, 56.
- (2) Cao, Y.; Smith, P.; Heeger, A. J. *Synth. Met.* **1992**, 48, 91.
- (3) Martin, C. R. *Science* **1994**, 266, 1961.
- (4) Liang, W.; Martin, C. R. *J. Am. Chem. Soc.* **1990**, 112, 8976.
- (5) Cai, Z.; Martin, C. R. *J. Am. Chem. Soc.* **1989**, 111, 4138.
- (6) Martin, C. R. *Adv. Mater.* **1991**, 3, 457.
- (7) Cai, Z.; Lei, J.; Liang, W.; Menon, V.; Martin, C. R. *Chem. Mater.* **1990**, 3, 960.
- (8) Wan, M. X.; Shen, Y. Q.; Huang, J. Chinese patent No. 98109916.5, 1998.
- (9) Huang, J.; Wan, M. X. *J. Polym. Sci., Part A: Polym. Chem.* **1999**, 37, 151.
- (10) Liu, J.; Wan, M. X. *J. Mater. Chem.* **2001**, 11, 404.
- (11) Liu, J.; Wan, M. X. *J. Polym. Sci., Part A: Polym. Chem.* **2001**, 39, 997.
- (12) Wan, M. X.; Huang, J.; Shen, Y. Q. *Synth. Met.* **1999**, 101, 708.
- (13) Shen, Y. Q.; Wan, M. X. *J. Polym. Sci., Part A: Polym. Chem.* **1999**, 37, 1443.
- (14) Wan, M. X.; Li, J. C. *J. Polym. Sci., Part A: Polym. Chem.* **1999**, 37, 4605.
- (15) Wan, M. X.; Liu, J.; Qiu, H. J.; Li, J. C. *Synth. Met.* **2001**, 119, 71.
- (16) Qiu, H. J.; Wan, M. X.; Matthews, B.; Dai, L. M. *Macromolecules* **2001**, 34, 675.
- (17) Qiu, H. J.; Wan, M. X. *Chinese J. Polym. Sci.* **2001**, 19, 65.
- (18) Yang, Y. S.; Wan, M. X. *J. Mater. Chem.*, in press.
- (19) Wan, M. X.; Li, J. C. *J. Polym. Sci., Part A: Polym. Chem.* **2000**, 38, 2359.
- (20) Funrhop, J. H.; Helfrich, W. *Chem. Rev.* **1993**, 93, 1565.
- (21) Elia, H. G. In *Macromolecules 2, Synthesis and Materials*; Plenum Press: New York, 1977; p 73.
- (22) Kim, B. J.; Oh, S. G.; Han, M. G.; Im, S. S. *Synth. Met.* **2001**, 122, 297.
- (23) Kim, B. J.; Oh, S. G.; Han, M. G.; Im, S. S. *Langmuir* **2000**, 16, 5841.
- (24) Harada, M.; Adachi, M. *Adv. Mater.* **2000**, 12, 839.
- (25) Huang, J.; Wan, M. X. *J. Polym. Sci., Part A: Polym. Chem.* **1999**, 37, 151.
- (26) MacDiarmid, A. G.; Chiang, J. C.; Halpern, M.; Huang, W. S. *Mol. Cryst. Liq. Cryst.* **1985**, 121, 173.
- (27) MacDiarmid, A. G.; Epstein, A. J. *Synth. Met.* **1994**, 65, 103.
- (28) Cao Y.; Smith, P.; Heeger, A. J. *Synth. Met.* **1989**, 32, 236.
- (29) Moon, Y. B.; Cao, Y.; Smith, P.; Heeger, A. J. *Polym. Commun.* **1989**, 30, 196.
- (30) Zhang, Z. M.; Wan, M. X. *Synth. Met.*, in press.
- (31) The P–O band length was calculated by the software: SymApps 5.0 written by Karl Nedwed, Copyright-1997 Bio-Rad Laboratories.
- (32) Mott, N. F.; Davis, E. A. *Electronic Processes in Non-crystalline Materials*, 2nd ed.; Clarendon Press: Oxford, 1979; p 34.
- (33) Kobayashi, A.; Xu, X.; Ishikawa, H.; Satch, M. *J. Appl. Phys.* **1972**, 5702.

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